

# Effect of Thermal Neutron Irradiation on the Growth and Structure of $K_2SO_4$ Crystal

T.A.AL-Dhahir Tagreed M.Al-Saadi

Department of Physics, College of Education for Pure Science, IbnAl- Haitham, Baghdad University ,Iraq

Email: Tariqaa52@yahoo.com;Tagreedmarush2000@yahoo.com

## Abstract

In this work we studied the effects of  $K_2SO_4$  crystal which is exposure to thermal neutron radiation. This crystal was grown by slow evaporation of the aqueous solution technique to the size  $(28 \times 6 \times 6) \text{ mm}^3$ . X-ray powder diffraction was used to identify the structures for before and after irradiation samples. Lattice parameter calculations revealed orthorhombic crystal system for before irradiation and monoclinic for after irradiation sample. Then we followed it by DSC which allowed us to highlight the high temperature phase transition around  $T_c 581.9^\circ\text{C}$  for before irradiation and at  $670^\circ\text{C}$  for after irradiation. The difference in  $T_c$  at high temperature during heating run indicates for structural changed. Then IR study verifies the reduction in crystal symmetry, which indicates to the correlation between thermal neutron irradiation and structure of  $K_2SO_4$ .

**Keywords:** Thermal neutron, Isomorphism, Single crystal, Phase transition, Symmetry reduction, after irradiation.

## Introduction

Potassium sulphate (PS)  $K_2SO_4$  is a material of great interest fertilizer and many other applications, like analytical chemistry as reagent, medicine, gypsum cements, glass manufactures, food additive (Periasamy et al 2009) and recently in biodiesel production Lipase-coated  $K_2SO_4$  micro-crystals (Zhengetal2012)[1,2]. The isomorphism structure of  $K_2SO_4$  crystals with  $K_2SeO_4$  salt compounds are the reference ones while studying the effects of various kinds in ferroic crystals of group. There are two groups of compounds with an  $A_2BX_4$ -type chemical formula having different crystal structures. One is the group of substances having an orthorhombic structure with a space group of  $P_{mnc}$  at room temperature. The other is the group having a monoclinic structure with a space group of  $P2_1/m$  at room temperature (Fuminao et al 2011)[3].  $K_2SO_4$  crystallizes in the room temperature into orthorhombic system with space group  $P_{mnc}(D_{2h}^{16})$  of lattice parameters  $a=5.763\text{\AA}$ ,  $b=10.071\text{\AA}$  and  $c=7.476\text{\AA}$  (Stadnyk et al 2013)[4]. It revealed a sequence of phase transitions that include disordered and incommensurate phases upon heating at  $587^\circ\text{C}$  into a hexagonal structure (commensurate ferroelastic phase) with lattice parameters  $a=5.921\text{\AA}$  and  $c=8.182\text{\AA}$ , of space group  $P6_3/mmc(D_{6h}^4)$  (Arnold et al 1981) [5]. Upon cooling, there is probably another phase transformation, from orthorhombic to monoclinic at 56 K with space group  $P2_1/b(C_{2h}^5)$  of displacive type. (Ahmed 1996)[6].

According to nuclear theory, thermal neutron may be captured by nuclei in the irradiated material, possibly changing its identity, produces an isotope of that nucleus, that is due to focus its energy into highly localized areas and the crystal structure of the irradiated material may be modified. The most common interaction of free neutrons with the nuclei of other materials are absorption and scattering. Scattering results in the transfer of some of its energy and the neutron continues to move through the irradiated material but at a lower energy and hence lower velocity. Its velocity is equivalent to thermal motion of a material, 2200 m/sec. and the corresponding energy  $\langle E \rangle$  is 0.025 eV, which is the most probable energy in the distribution at room temperature ( $20^\circ\text{C}$ ). The average energy of thermal neutrons at room temperature is 0.038 eV. Which is the upper limit  $\sim 0.4$  eV (Cadmium cutoff) and epithermal, just above 0.4 eV. As thermal neutrons are of low energy, usually only a high flux density or a long exposure induces the change in materials (James 2007) and (Myrzakhmet 20013)[7,8]. Kamali and Walton 1985 [9], attributed their electron spin resonance spectra of (PS), which was irradiated by gamma, electron, neutron and fission fragments due to the formation of  $SO_3^-$ ,  $SO_4^-$ ,  $SO_2^-$ , and  $O_3^-$  radical ions. The most intense radical ion was due to the  $SO_3^-$ , and the other radicals were relatively much lower in intensity. There have been many articles published, concerning both pure and applied aspects, on the;  $K_2SO_4$  however, the use of thermal neutron radiation to modify the properties of (PS) is relatively unreported. The present work focuses on irradiated (PS) by thermal neutron, grown single crystal to the size of  $(28 \times 6 \times 6) \text{ mm}^3$  then study the effect of thermal neutron irradiation on its properties. So, the Fundamental goals of irradiation are to correlate physical properties of (PS) with their crystal structures and to determine the impact of irradiation on crystal parameters and phase transitions.

## 2 Experimental details

### 2-1 Growth of the crystal

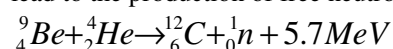
Single crystals of  $K_2SO_4$  was grown by using slow evaporation technique from saturated aqueous solution of one molar concentration. The solution prepared by dissolving 34.86 gm of potassium sulphate powder, (Chem-

Supply PTY.LTD analytical reagent AR 99.5%) as a source of (PS) which was purified by re crystallization into 200 ml of double distilled water .The aqueous solution was mixed with magnetic stirrer for two hours at 200 rpm and 50° C . Then, we poured it into crystallization vessel, during slow evaporation process; the crystallizers were covered with perforated plastic sheet and kept in a dust free in the undisturbed oven at 40°C .After 3-5 days, small crystals were grown as a result of spontaneous nucleation in the bottom of the vessel. It appeared as singular and of regular shape with well faceted .So, we allowed it to grow larger. At the end of growth period ( 50 days) a crystal of dimensions( 28 x 6x 6 )mm<sup>3</sup> are obtained. Some of the grown crystal was exposed to neutron irradiation. Crystal habit for the grown crystal and its morphology will be discussed.

## 2-2 Sample preparation for irradiation

Powder samples are obtained from the grown crystal, then divided into two parts. One of them was kept as reference specimen and the other was irradiated.

The neutron radiation source used for the sample irradiation was produced from an<sup>241</sup>Am-<sup>9</sup>Be source with activity 12Ci and flux10<sup>5</sup>n.cm<sup>-2</sup>.s<sup>-1</sup>. The sources is made by mixing the target material like Beryllium with a suitably strong  $\alpha$ - source. When Beryllium is bombarded by  $\alpha$  particle, it undergoes a number of reactions, which lead to the production of free neutrons according to ( $\alpha$  n) type nuclear reaction).



Taking Am<sup>241</sup>as an  $\alpha$ -source of energy to yield a neutrons beam of (6.048x10<sup>10</sup> n.cm, that is used to irradiate the sample for seven days in room temperature and atmospheric pressure.

## 2-3 Powder XRD

The X-ray diffraction patterns were recorded in a (Shimadzu XRD-6000) using copper target (Cu K $\alpha$ , 1.5418 Å), (40 kV and 30 mA). The samples were mounted an on glass sample holder. Step-scan data were collected from different ranges with a step width of 0.02° and a counting time of 5 sec/step.

3-4The mid-infrared,

FTIR Shimadzu spectrophotometer model 8400S is used to study the fundamental vibrations K<sub>2</sub>SO<sub>4</sub> .The (PS) samples were finely ground for one minute, combined with oven dried KBr (2 mg K<sub>2</sub>SO<sub>4</sub>/200 mg spectroscopic grade KBr) and pressed into a disc .The spectrum of the sample was recorded in transmittance (%) mode with 50 scans at 4 cm<sup>-1</sup> resolution between 400 cm<sup>-1</sup> and 4000 cm<sup>-1</sup> (2.5–25  $\mu$ m),for pre- irradiated and post irradiated samples at room temperature.

4-5Differential scanning calorimetric is preformed using Linsess (STR PT-1000)

## 3 Results and Discussion

### 3-1Crystal growth and morphology

Transparent, colorless single crystals of (PS) were grown from an aqueous solution by slow evaporation for before irradiation sample .The photograph of the as-grown crystals is presented in Figure (1).

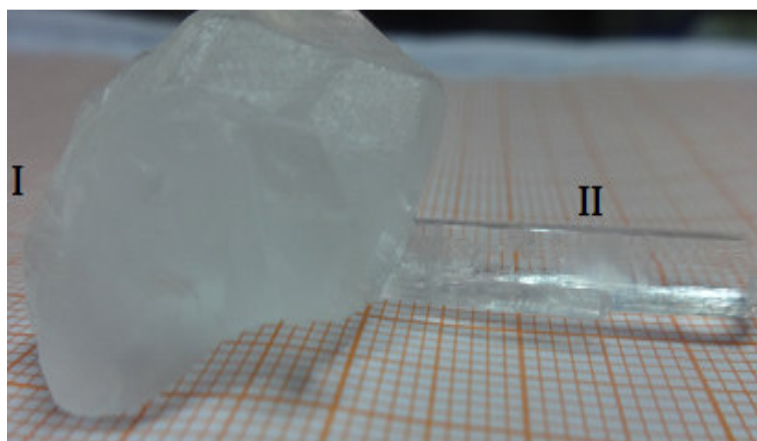


Fig.(1)Photography of the grown K<sub>2</sub>SO<sub>4</sub> crystal ,(I)Dipyramidal, and (II) Elongated hexagonal pyramids.

K<sub>2</sub>SO<sub>4</sub> is crystallized in two different habits of crystal belonging to orthorhombic .This is due to the effect of interactions between solute and solvents molecules at the various crystal interfaces, which may have a dominant role on the shape of the crystal morphology that is obtained from solution growth .So, preferential adsorption of solvent molecules on a particular faces delay the growth rate of other faces(Singh 2006)[10].Left side( I )shows a crystal a predominant dipyramidal or prismatic habit with dimensions (25x22x18)mm<sup>3</sup>, with b-

axis along the long axis of the pyramid and the c-axis was along one edge of the quasi-triangular basal plane. While the other one is on the right side with dimensions ( 28 x 6x 6 ) mm<sup>3</sup> , represent a combination crystal terminated by hexagonal pyramids from one end and with six flat parallel faces along the direction of its elongation which was with the same form as that presented by (Mullin 2001)([11]). Twinning is occurred in(PS) crystal grown from aqueous solution with three ferroelastic domains under a polarizing microscope at room temperature, and their boundary(domain walls) disappeared at 860K(Chbn and Wang 1997)[12].This means triple twin is related to three twin mirror planes or twofold twin axis .The permissible walls for any two adjacent domain pairs are derived by the symmetry approach to a crystal undergoing the structural phase transition by considering the twin structures across the domain boundaries (Dhanaraj et al 2010)[13].

Fig1 exhibits the habit of actual shape of the as grown single crystal .The difference in the two grown crystal form ( I &II) appeared due to the possible of non uniformity of actual conditions of growth, like temperature gradient, concentrate and other factor. So, the actual shape does not differ drastically from the ideal. The re crystallization growth of after irradiation (PS) K<sub>2</sub>SO<sub>4</sub> sample ,lead to prismatic habit to the dimensions (13x11x5) mm<sup>3</sup> with a flat plane as shown in fig.2a.While fig .b represents the photograph crystal (II) with more details about its planes and direction. Fig.c represents a single part from other side of part(II) with marked Miller indices for the planes of the grown K<sub>2</sub>SO<sub>4</sub> crystal.

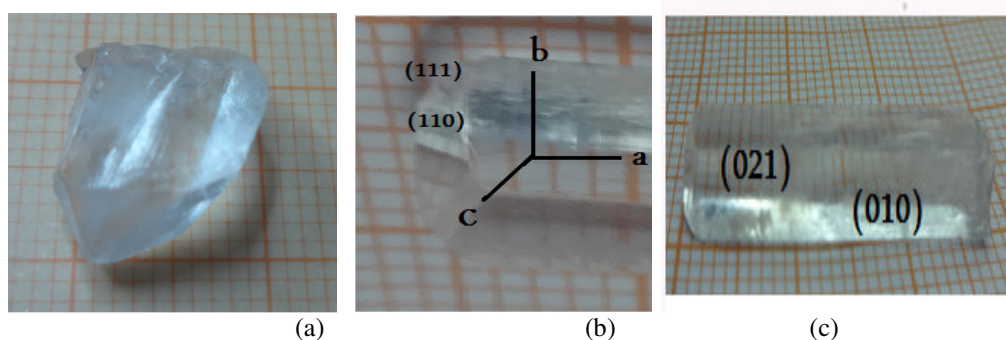
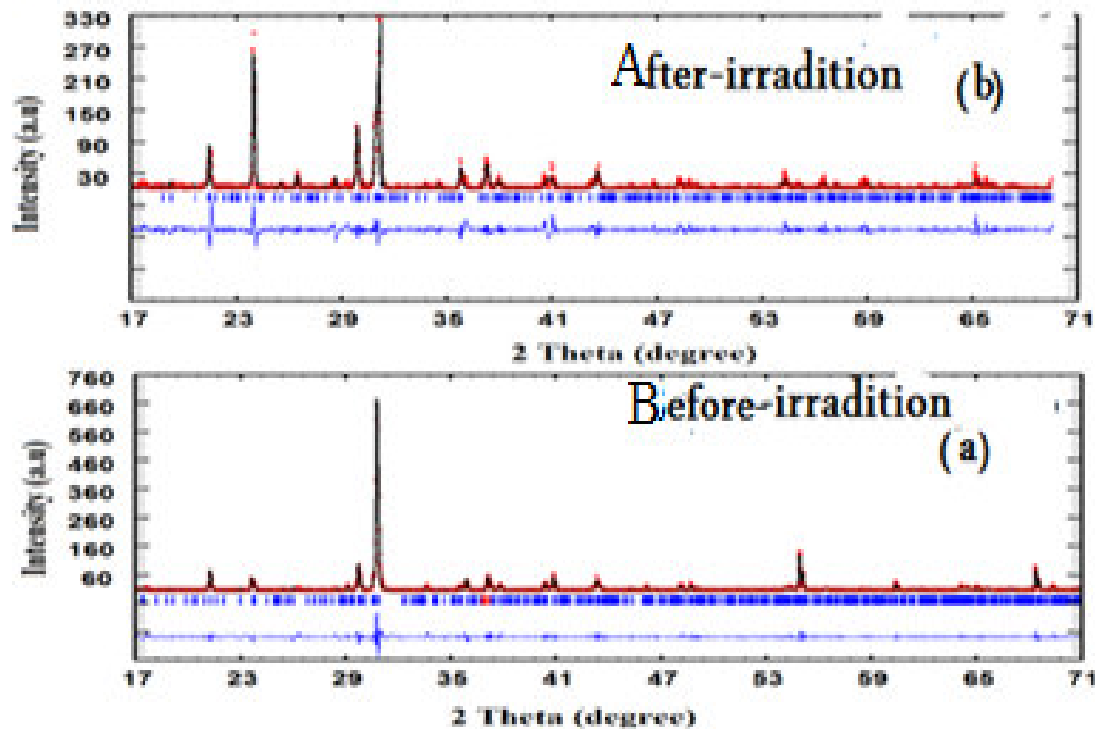


Fig.2. photograph grown K<sub>2</sub>SO<sub>4</sub> crystal, a- prismatic habit for after irradiation growth, b &c for before irradiation crystal growth .

### 3-2 X-RAY diffraction analysis

Powder x-ray diffraction studies were performed on the powder of the grown crystals to identify the phase formation and degree of crystal perfection. Its pattern of before irradiation and after irradiation by thermal neutron was recorded. Figure 3a shows the diffraction peaks of before irradiation matched well with peaks that reported for (PS) crystal in the literature((Fuminao et al 2011), (Stadnyk et al 2013) ( Arnold et al 1981 )and (Ahmed 1996) [3-6].That is due to the clear crystalline samples, so, the occurrence of sharp peaks at specific Bragg's angles. The recorded data were treated by Rietveld analysis with "full prof"(fullprof,VersionApril2014-ILL JRC(Roisnel and Rodriguez 2014)[14] shows the calculated lattice parameters are in good agreement with the reported values. It is clear from figure 3b, that the after irradiation (PS) K<sub>2</sub>SO<sub>4</sub> crystals lead to a change in the intensity, shift of peaks when compared with that peaks of before irradiation. So, the irradiation has brought a change in the internal structure of crystals due to change in bond lengths or the angle. According to the data analysis of the XRD patterns presented in figure 3a, the lattice parameter  $\alpha = \gamma = 90^\circ$  and  $\beta = 90.5415^\circ$ , and  $a(\text{\AA}) = 10.01577$ ,  $b(\text{\AA}) = 7.47461$  and  $c(\text{\AA}) = 5.78989$  with conventional unit cell volume  $V = 433.43\text{\AA}^3$  are obtained. It is clear that, the angle beta ( $\beta$ ) is not 90, this is due to the twinning of the crystal. So the possible pitfalls errors are in the space-group determination outcome. This is occurred to monoclinic crystal (b unique) with the angle fortuitously equal to  $\sim 90^\circ$  ( in our case  $\beta = 90.5415^\circ$  ).Since there is twinning according to (100)plane , this will cause overlap of the reflections  $hkl$  and  $\bar{h}kl$ . So, the observed Laue symmetry is mmm (orthorhombic) crystal system rather than 2/m (monoclinic)( Theo 2005 )[15] . So ,it is possible to be pseudo orthorhombic.



Figure(3);X-ray diffraction pattern for  $K_2SO_4$ (a)before irradiation,(b)after irradiation

While lattice parameters for the after irradiation sample are calculated from the data of fig.3b which revealed the crystals belong to monoclinic system as  $\alpha = \gamma = 90^\circ$  and  $\beta = 101.7997^\circ$ ,  $a(\text{\AA}) = 10.0842$ ,  $b(\text{\AA}) = 7.5810$  and  $c(\text{\AA}) = 4.6398$  of space group with conventional unit cell volume  $V = 344.45 \text{\AA}^3$ . It is clear from the comparison of the unit cell calculated parameters of before and the after irradiation, that the after irradiation sample is with monoclinic crystal system rather than pseudo orthorhombic; and the volume of unit cell is reduced by 21% from that of before irradiation. In literature there are two groups of halide compounds with an  $A_2BX_4$ -type chemical formula having different structures. One is the group of substances of orthorhombic  $\beta$ - $K_2SO_4$  structure with a space group of  $P_{mcn}$  at room temperature. The other is the group having a monoclinic like  $Sr_2GeS_4$  structure with a space group of  $P2_1/m$  at room temperature. Furthermore,  $Cs_2CdI_4$  and  $Cs_2HgI_4$  transform from the monoclinic  $Sr_2GeS_4$  structure (called the  $\alpha$ -phase) to the orthorhombic  $\beta$ - $K_2SO_4$  structure (called the  $\beta$ -phase) at high temperatures, and this phase transition is called an  $\alpha$ - $\beta$  phase transition (Fuminao et al 2011)[3]. So, thermal neutron irradiation induces a lattice change and there is possibility of (PS) crystal structure changed from pseudo orthorhombic before irradiation to monoclinic crystal system of the same as the structure of  $Sr_2GeS_4$ .

$K_2SO_4$  revealed a ferroelastic phase transition temperature at ( $T_c = 584^\circ\text{C}$ ) with symmetry change according to Aizu notation  $6/mmm$  to  $mmm$ . The hexagonal structure of space group  $D_{6h}^4 = P6_3/mmc$  (Park 1998 ).[16]

### 3-3 Differential Scanning Calorimetric

To compare the results of DSC measurements of heating and cooling run with rates of  $\pm 5\text{K/min}$  around  $T_c$  for both samples of the grown (PS) crystals with that published data (Anooz et al 2008 ) [17]. The heating run for before irradiation revealed the expected endothermic peak maximum  $T_c$  at  $581.9^\circ\text{C}$ , while it revealed an exothermic  $T_c$  at  $577.9^\circ\text{C}$  after cooling the sample. There is hysteresis of about  $4^\circ\text{C}$ . This behavior matched very well to the high-temperature hexagonal phase of (PS) (Park 1998 ) [16].

While for that for after irradiation sample,  $T_c$  is appeared at  $670^\circ\text{C}$  for heating and  $566^\circ\text{C}$  for cooling run (with hysteresis of about  $104^\circ\text{C}$ ) (AL-Dhahir) [18]. The hysteresis arises for first-order phase transitions because nucleation of the new phase is activated. The change free-energy positive until the nucleus reaches a critical size beyond which further growth is downhill. The transformation at  $670^\circ\text{C}$  leads us to suggest that crystal system of this phase is either hexagonal or orthorhombic. The result of DSC for after irradiation sample encourages us to go to Aizu notation (Aizu 1970) [19] and look for a species which is matching with this

transformation. Since ferroelastic hexagonal is the crystal system of high temperature phase as reported earlier (Park 1998) [16], and our room temperature structure is monoclinic as mentioned from the x-ray structure analysis. So, Aizu notation for this phase transition may be  $6/mmmF2/m$ , or  $mmmF2/m$ . Details will be considered in future studies for ferroelatic and the crystal structure for irradiated at high temperature.

### 3-4 Fourier transform Infrared Transmission Spectroscopy

The FTIR spectroscopy studies are used to analyze qualitatively the presence of functional groups in the grown (PS) crystals for before thermal neutron irradiation.

Fig.(4a) presented the absorption bands in the region of  $3485.37\text{ cm}^{-1}$  and  $3415.93\text{ cm}^{-1}$  are corresponding to O-H stretching vibrations of water and hydroxyls. These are due to the symmetric and asymmetric stretching of the O-H bond. The peak observed at  $3485.37\text{ cm}^{-1}$  is due to asymmetric O-H stretching vibration present in the compound. According to the selection rules for  $T_d$  symmetry, the IR spectrum of  $\text{SO}_4^{2-}$  ion has two absorption bands  $\nu_3$  which correspond the asymmetric stretching vibrations of the S-O bonds, and  $\nu_4$  which correspond to the bending vibrations of O-S-O. Each of these frequencies are triply degenerated. These two bands triply degenerated vibrations ( $\nu_3$  and  $\nu_4$ ) are observed in the IR absorption spectrum of at (PS)  $1114.86\text{ cm}^{-1}$  and  $617.22\text{ cm}^{-1}$ . The strong peaks observed at  $1114.86\text{ cm}^{-1}$  and weak at  $1160\text{ cm}^{-1}$  are attributed to triply degenerate of asymmetric stretching mode ( $\nu_3$ ) of  $\text{SO}_4^{2-}$ . Since the symmetry of  $\text{SO}_4^{2-}$  in solution is nearly tetrahedral (Ilamaran 2012) and (Satish 1998) [20,21], with a totally symmetric S-O stretching vibration at  $980\text{ cm}^{-1}$  ( $\nu_1$ , IR inactive) and an asymmetric, triply degenerate, S-O stretching vibration at  $1102\text{ cm}^{-1}$  ( $\nu_3$ , IR active). The deformation and combination vibrations ( $\nu_2$  and  $\nu_4$ ) for S-O occur at lower energy. If the tetrahedral  $\text{SO}_4^{2-}$  symmetry is lowered, their IR spectrum changes and becomes IR-active. The triple degeneracy of  $\nu_3$  is lifted, which results in a split of the  $\nu_3$  band into two or three bands depending on the symmetry of the structure formed. The spectral changes are on coordination of the sulfate anion. Therefore, the degenerate due to  $T_d$  tetrahedral deformation leads to symmetry reduction to  $C_{3v}$  (Hug 1997) [22]. The single strong peak at  $617.22\text{ cm}^{-1}$  assigned as asymmetric bending mode of  $\text{SO}_4^{2-}$ , a weak sharp peak appears at  $981.77\text{ cm}^{-1}$  ( $\nu_1$ ), corresponding to the symmetric stretching vibration of (S-O). The band observed at  $482.20\text{ cm}^{-1}$  is assigned to the doubly degenerate ( $\nu_2$ )  $\text{SO}_4^{2-}$ .

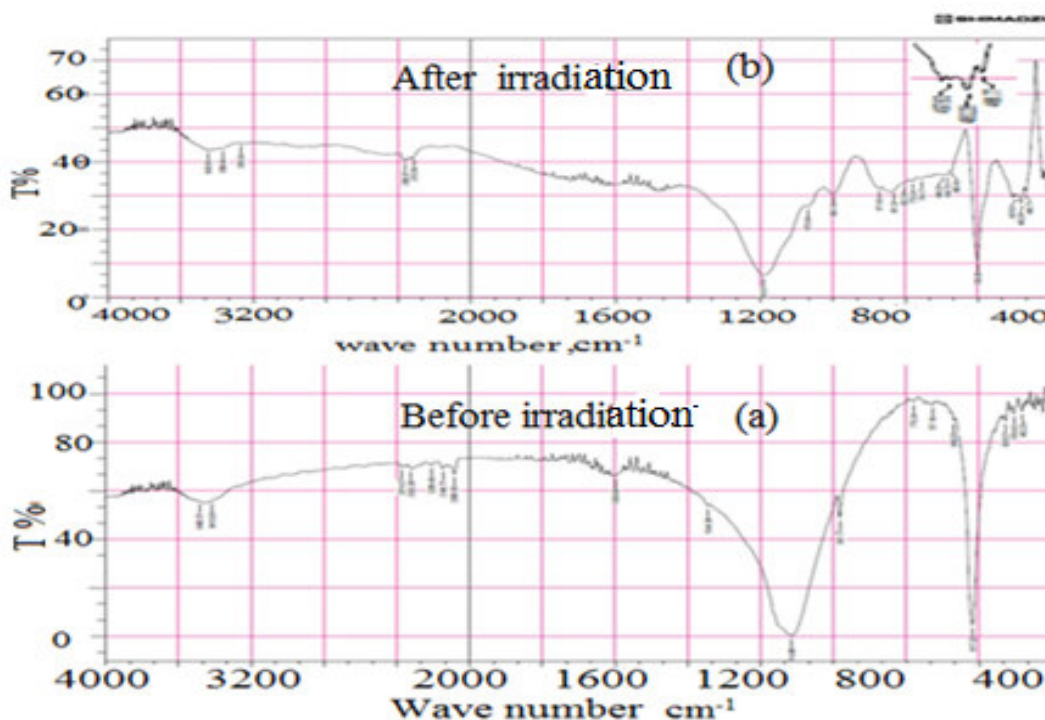


Fig.4:Fourier transform Infrared Transmission ,a-before thermal neutron irradiation, b-after neutron irradiation.

The FTIR spectrum for the after thermal neutron irradiation is presented in fig.(4b). It showed an absorption band in the region of  $3450.65\text{ cm}^{-1}$  and  $3369.64\text{ cm}^{-1}$  which are corresponding to OH stretching vibrations of water and hydroxyls, and are due to the symmetric and asymmetric stretching of the O-H bond both are shifted to lower wave number. The bands appeared at  $1192.01\text{ cm}^{-1}$  and  $1070.49\text{ cm}^{-1}$  shifted to higher wave number than that appeared for before irradiation and assigned to asymmetric stretching sample. Forbidden, but occasionally be seen as a very weak band at  $1000\text{ cm}^{-1}$  (Periasamy et al 2009) [1]. So it is observed at

$999.13\text{cm}^{-1}$ . Distortion of tetrahedral due to electrostatic effects, could shift the  $\nu_3$  to higher wave number and cause the  $\nu_1$  band to become IR active (Derek et al 1999)[23]. If sulfate forms a bidentate binuclear (bridging) surface complex, due to the irradiation by thermal neutron the symmetry is further lowered to  $C_{2v}$ , then the  $\nu_3$  band splits into three bands between  $1070.49$  and  $1192.01\text{cm}^{-1}$ , while the  $\nu_1$  band is shifted to around  $1000\text{cm}^{-1}$ . As the non degenerate symmetric stretching  $\nu_1$  and the triply degenerate asymmetric stretching  $\nu_3$  bands are the two infrared sulfate vibrations  $\nu_1$  and  $\nu_3$  that are accessible to spectroscopic investigation (Nakamoto 1986) [24]. The bands appeared below  $700\text{cm}^{-1}$  are assigned to the symmetric and asymmetric bending ( $\nu_2$ ) O-S-O of  $\text{SO}_4$  groups. Hence we have assigned the frequency  $599.86\text{cm}^{-1}$  to the asymmetric bending, and  $497.83$ ,  $480.28$  and  $466.77\text{cm}^{-1}$  to symmetric bending of the  $\text{SO}_4$  groups. The band observed at  $669.30\text{cm}^{-1}$  is assigned as asymmetric bending. The non degenerate symmetric stretching  $\nu_1$  and the triply degenerate asymmetric stretching  $\nu_3$  bands are two infrared sulfate vibrations that are accessible to spectroscopic investigation.

#### 4-Conclusion

In conclusion, the stable structure of  $\text{K}_2\text{SO}_4$  at room temperature is orthorhombic for before thermal neutron irradiation and monoclinic structure for after irradiation. Both samples showed a high temperature first order phase transition with different  $T_c$  and different hysteresis range. Fourier transform infrared transmission spectroscopy carried at room temperature revealed a reduction in symmetry for after irradiation sample. Unfortunately, it is not clear the connection between room temperature suggested monoclinic structure phase for after irradiation sample due to unavailability of facility at high temperature X-ray scattering. So, Further structural study is needed.

#### Acknowledgments

The author wants to express his gratitude to Prof. Dr Khalid H. Mahdi for thermal neutron irradiation facility.

#### References

- [1] Periasamy Azha, Muruganand, S. and Palaniswamy M. (2009). Vibrational studies of  $\text{Na}_2\text{SO}_4$ ,  $\text{K}_2\text{SO}_4$ ,  $\text{NaHSO}_4$  AND  $\text{KHSO}_4$  crystal. Rasyan. j.Chem. 2(4) 981-989
- [2] Zheng, J. Liu, Xu L, Zhang, Y. Yan, X (2012). Lipase-coated  $\text{K}_2\text{SO}_4$  micro-crystals: preparation, characterization, and application in biodiesel production using various oil feed stocks. Bioresour Technol. Apr; bior tech. Epub 2012 Jan 28.
- [3] Fuminao, S. Masaaki, T and Toshihisa, Y (2011). Time dependence of the  $\alpha$ - $\beta$  phase transition in  $\text{Rb}_2\text{CdBr}_4$ . Journal of the Korean Physical Society, 59(3) 2475-2477
- [4] Stadnyk, V.YO. Kashuba, O.Z. Brezvin, R.S. I.M. Matviishyn, I.M. M.Y. and Rudysh. M.Y. (2013). Birefringence properties of uniaxially compressed  $\text{K}_2\text{SO}_4$  crystals; Ukr. J. Phys. 58(9) 853-856
- [5] Arnold, H. Kurtz, W. Richter-Zinnius, A. Bethke, J. Heger, G (1981). The phase transition of  $\text{K}_2\text{SO}_4$  at about 850 K, Acta Cryst. B 37, 1643-1651
- [6] Ahmed, S.A (1996). X-ray studies of the low-temperature phase of  $\text{K}_2\text{SO}_4$ . Physica Status Solidi (b) 195 (1) 113-118.
- [7] James E. Turner (2007). Atoms, Radiation, and Radiation Protection, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim. (chapter 9)
- [8] Myrzakhmet, M. Nikl, M. Jary, V. Datelkhan, B. Zhanabergenov A. (2013). Thallium-doped sulphate potassium crystals as materials for radiation detectors. Functional materials 20(3) 295-299.
- [9] Kamali, J. and Walton, G.N. (1985) Electron spin resonance of gamma, electron, neutron and fission fragments irradiated  $\text{K}_2\text{SO}_4$ ; Radiation Effects; 84, 171-184.
- [10] Singh M. K. (2006) First principle study of crystal growth morphology: An application to crystalline urea; Condensed Matter, Materials Science, rXiv: cond-mat/0602385v2.
- [11] Mullin, J. W. (2001). Crystallization. Fourth editions. Butterworth Heinemann. (Chapter 1).
- [12] Chbn, R. H. and Ren-June Wang (1997) Studies on structural phase transition and ferroelastic domain structures in  $\text{K}_2\text{SO}_4$  crystal. Phase Transitions: A Multinational Journal, 62(4) 199-207.
- [13] Dhanaraj, G. Byrappa, K. Prasad, V. Dudley, M. (2010); Springer Handbook of Crystal Growth, p124.
- [14] Roisnel, T and Rodriguez-Carvajal (2014) Winplotr, version April 2014.
- [15] Theo, H (2005) International tables for crystallography, Volume A: Space-Group Symmetry, Editor Theo Hahn, Fifth edition, Springer, p45.
- [16] Park, S.E. Jeong, H. T. Cho, Y. C. and Jeong, S. Y. (1998) Ferroelastic properties of  $\text{K}_2\text{SO}_4$  crystals, Journal of the Korean Physical Society, Vol. 32, No. , February, pp. S786-S788.
- [17] Anooz, S. B. Klimm, D. Schmidbauer, M. Bertramb, R. RoXberg, M. (2008); Effect of  $\text{Cd}^{+2}$  on the growth and thermal properties of  $\text{K}_2\text{SO}_4$  crystal, Journal of Physics and Chemistry of Solids 69, 2356-2359.
- [18] AL-Dhahir. T.A, it will published else where,

- [19] Aizu, K. (1970) Possible species of ferromagnetic, ferroelectric and ferroelastic crystals; *Physical Review B* 2 (3) 754-771.
- [20] Ilamaran, C. Sethuram, M. Dhandapani, M and Amirthaganes, G. (2012) Crystal growth, FTIR and thermal characterization of bis(ethyltriphenylphosphonium) tetrabromomanganate(II) dihydrate crystals, *Pramana Journal Physics*, 2012 Indian Academy of Sciences 78, (5) 791-801
- [21] Myneni, S.C.B. Samuel, J. T. Waychunas, G. A and Logan, T. (1998) Vibrational spectroscopy of functional group chemistry and arsenate coordination in ettringite; *Geochimica et Cosmochimica Acta*, 62(21/22), 3499-3514.
- [22] Hug, S. J. (1997) In Situ Fourier transform infrared measurements of sulfate adsorption on hematite in aqueous solutions; *Journal of Colloid and Interface Science* 188, 415-422.
- [23] Peak, D. Ford, R. G and Sparks, D. L. (1999) An in Situ ATR-FTIR investigation of sulfate bonding mechanisms on goethite. *Journal of Colloid and Interface Science* 218, 289-299.
- [24] Nakamoto, K. (1986) *Infrared and Raman spectra of inorganic and coordination compounds.* Wiley, New York.

The IISTE is a pioneer in the Open-Access hosting service and academic event management. The aim of the firm is Accelerating Global Knowledge Sharing.

More information about the firm can be found on the homepage:

<http://www.iiste.org>

## CALL FOR JOURNAL PAPERS

There are more than 30 peer-reviewed academic journals hosted under the hosting platform.

**Prospective authors of journals can find the submission instruction on the following page:** <http://www.iiste.org/journals/> All the journals articles are available online to the readers all over the world without financial, legal, or technical barriers other than those inseparable from gaining access to the internet itself. Paper version of the journals is also available upon request of readers and authors.

## MORE RESOURCES

Book publication information: <http://www.iiste.org/book/>

Academic conference: <http://www.iiste.org/conference/upcoming-conferences-call-for-paper/>

## IISTE Knowledge Sharing Partners

EBSCO, Index Copernicus, Ulrich's Periodicals Directory, JournalTOCS, PKP Open Archives Harvester, Bielefeld Academic Search Engine, Elektronische Zeitschriftenbibliothek EZB, Open J-Gate, OCLC WorldCat, Universe Digital Library, NewJour, Google Scholar

